THE PREPARATION, CHARACTERIZATION, AND THERMAL BEHAVIOUR OF SOME LITHIUM ALUMINUM OXIDES: Li₃AlO₃ AND Li₅AlO₄

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The purpose of this research was to study the best conditions for the synthesis of the double oxides Li_5AlO_4 and Li_3AlO_3 in the solid state starting from the simple oxides, and to determine their heats of formation.

 Li_5AlO_4 was obtained from Li_2O_2 or Li_2O and γ -Al₂O₃ in a Li/Al molar ratio of 5 : 1, and was characterized by X-ray methods. Lithium orthoaluminate, Li_3AlO_3 , was obtained from Li_2O_2 and γ -Al₂O₃ in a molar ratio 3 : 1. The postulated formula, Li_3AlO_3 , was confirmed by chemical analysis.

The temperature ranges in which the compounds are stable were established by the DTA method, and were found to be very limited for Li_3AlO_3 (400-430°) but greater for Li_5AlO_4 (440 - more than 600°).

The heats of formation of Li_5AlO_4 and Li_3AlO_3 , also determined by means of the DTA method, were found to be -552.3 ± 0.8 kcal/mole and -416.8 ± 2 kcal/mole, respectively.

The number and the stabilities of the compounds formed by reaction between Li_2O and Al_2O_3 are rather uncertain; it is well known that the possibility of the formation of lithium aluminum double oxides is highly dependent on the Li_2O/Al_2O_3 molar ratio and on the firing temperature. Debray and Hardy [1], Hagenmuller and Debray [2], Kordes [3], Lehmann and Hesselbarth [4], Zintl and Morawietz [5], and Hummel et al. [6] reported that the only high-temperature forms of lithium aluminum double oxides are h-LiAlO₂ (h = high-temperature modification) and LiAl₅O₈, whereas Lehmann and Hesselbarth [7], and Torkar et al. [8] concluded that 1-LiAlO₂ (low-temperature modification) and Li_5AlO_4 are the only existing low-temperature forms.

The purpose of the present work was to investigate in some detail the reactivity conditions in the solid state between aluminum oxides and lithium compounds in order to obtain pure Li-Al-O compounds and to determine the possibility of the formation of other so far unknown mixed oxides.

In particular we investigated mixtures with Li/Al molar ratios of 5:1 and 3:1 (at different firing temperatures), in order to obtain pure Li₅AlO₄ and to examine its thermal behaviour; moreover, our aim was to discover the formation conditions of other double oxides such as Li₃AlO₃, a compound already postulated by Kröger and Fingas [9] and by Fedorov and Shamrai [10], but never characterized in its chemical and physical properties, and also to examine its thermal behaviour.

Experimental and results

Preparation

 γ -Al₂O₃, prepared by calcination at 500° of the precipitate formed from Al(NO₃)₃ in ammonia solution, was used as the aluminum reagent. We preferred to use γ -Al₂O₃, instead of the α -Al₂O₃ used by several other authors, because of its higher surface area and thus its higher reactivity in the solid state; γ -Al₂O₃ was therefore a fixed component for all the preparations.

For the lithium reagent, lithium nitrate and carbonate were used in the first stage of the investigation, as suggested by several authors; however, probably due to the high temperature of decomposition of $LiNO_3$ and Li_2CO_3 to give Li_2O [11], the above lithium compounds did not react very easily with Al_2O_3 (many unwanted phases were observed by X-ray powder analysis). It was thus thought preferable to use highly pure Li_2O and Li_2O_2 (supplied by "LCAO" Co.) as starting materials. The reagents (Al_2O_3 and Li_2O or Li_2O_2) were mixed in a mortar in Li/Al molar ratios from 3 : 1 to 5 : 1 and each mixture was calcined in the temperature range $400-700^\circ$ for a number of hours varying from 10 to 500. After each firing an X-ray powder spectrum was prepared for an examination of the phases present in the samples.

It was found that for the Li_5AlO_4 preparation the best firing temperature is about 500° in zir. At temperatures lower than 500°, phases such as Li_2O and Al_2O_3 are still present, whereas at higher temperatures (600-700°) the presence of 1-LiAlO₂, Li_2O and Li_5AlO_4 is observed.

The compound was finally obtained completely pure as a fine, white powder under the following conditions: $\text{Li}_2\text{O}_2 + \gamma - \text{Al}_2\text{O}_3$ in the molar ratio 5:1 at 500° in air for 140 hours. The formation of Li_5AlO_4 was also observed in the reaction of $\text{Li}_2\text{O} + \text{Al}_2\text{O}_3$ (5:1) at 500°, in agreement with the findings of Lehmann and Hesselbarth [7]. It should be noted that when Li_2CO_3 or LiNO₃ were used as starting materials, a certain amount of 1-LiAlO₂ was also obtained at 500°.

For the preparation of the lithium orthoaluminate, Li_3AlO_3 , as described in detail later, the best conditions were to heat a 3 : 1 $Li_2O_2 - Al_2O_3$ mixture at 400° in air for 500 hours.

An attempt to prepare this compound from $Li_2O + Al_2O_3$ (at the same firing temperature, 400°, and a molar ratio 3 : 1) was unsuccessful. Using these reagents we obtained only LiAlO₂ (l-form) and Li_5AlO_4 phases.

X-ray spectra

Phases were identified by a Philips X-ray diffraction instrument using CuK_{α} (nickel-filtered) radiation. The X-ray patterns were taken both with a Debye-

Table 1

		Our data		Lehmann and Hesselbarth [7		
No.	20°	d	ja	d	I	
1	17:2	5 16	137	5.21	w	
2	19.8	4.48	vvw	4 48	vvw	
3	22.0	4.04	s	4 07	s	
4	23.2	3.83	NW NW			
5	24.0	3 71	vs	3 74	vs	
6				3 38	vvw	
7	29.8	3.00	vvw	3.02	vvw	
8	30.8	2.90	vvw		_	
9	32.0	2.80	vvw	_	_	
10	32.8	2.73	vvw	2.75	vvw	
11	33.3	2.69	vw		_	
12	34.5	2.60	vs	2.62	vs	
13	35.8	2.51	vvw	2.52	vvw	
14	36.1	2.49	vvw			
15	37.0	2.43	vw	2.43	vw	
16	37.5	2.40	vvw	_	_	
17	38.1	2.36	vvw	_		
18	39.0	2.31	vvw	_	_	
19	40.5	2.23	vvw	2.26	vvw	
20	42.2	2.14	vvw	2.15	vvw	
21	43.5	2.08	vvw	2.08	vvw	
22	46.0	1.97	vvw	1.98	vvw	
23	46,5	1.95	vvw		_	
24	47,4	1.92	w	1.93	w	
25	49.3	1.85	vvw	1.87	vvw	
26	50.5	1.81	vvw		_	
27	51.4	1.78	vw	1.78	vw	
28	52.7	1.74	vvw	1.75	vvw	
29	54.5	1.68	vw	1.69	VW	
30	55.5	1.66	vvw	1.66	vvw	
31	57.4	1.61	vvw	_	_	
32	57.9	1.59	s	1.61	s	
33	59.0	1.57	vvw	_	_	
34	60.4	1.53	vvw	1.53 ^b	vvw	
35	61.1	1.52	vvw			
36	63.3	1.47	vvw	1.47 ^b	vw	
37	63.8	1.46	vvw			
38	67.4	1.39	vvw	1.40	vvw	
39	68.5	1.37	vw	1.38	vw	
40	69.5	1.35	vw	1.35 ^b	w	
41	70.3	1.34	vw			
42	73.0	1.30	vvw	1.30	vvw	
43	77.2	1.24	vw	1.24	vw	
44	78.5	1.22	vvw	1.23	vvw	
45	81.3	1.18	vvw	1.19	vvw	
	1					

X-ray powder pattern of $\text{Li}_{5}\text{AlO}_{4}$

	0	ur data		Lehmann and Hesselbarth [7]		
No,	20°	d	Ia	d	I	
46	84.0	1.15	vw	1.16	VW	
47	85.2	1.14	vw	1.14	vw	
48	87.3	1.12	vw	1.12	vw	
49	89.7	1.09	vvw	1.10	vvw	

Table 1 (cont.)

^a Visually estimated intensities: vs = very strong; s = strong;

w = weak; vw = very weak; vvw = very very weak.

^b Broad diffuse bands, as observed by the authors of [7].

Scherrer powder photographic camera, 114.6 mm in diameter (Straumanis mounting technique), and with a Philips powder goniometer using an angular speed of $1/4^{\circ}$ (2 Θ) per minute. The lines on the X-ray films were read visually with a Norelco measuring device and their positions corrected for film shrinkage.

Table 1 reports the values of the Bragg 2Θ angles, the interplanar *d* spacings, and the estimated intensities *I* for the X-ray spectrum of the sample obtained by calcining Li_2O_2 (or Li_2O) + Al_2O_3 (5 : 1) at 500° for 140 hours, i.e. of Li_5AlO_4 . In the same Table the data reported by Lehmann and Hesselbarth [7] for Li_5AlO_4 are presented for comparison. In this product no other phases were detected, indicating that the above preparation gives a pure, uncontaminated Li_5AlO_4 compound. It should be noted that we have observed 12 very very weak reflections not reported by Lehmann and Hesselbarth: since these peaks could not be attributed to any phase other than Li_5AlO_4 we think that this difference in the spectra is probably due to a longer film exposure time applied by us. As regards the peak at d = 3.38 Å which was observed (very very weak) by L. and H. [7] we did not detect it at all.

Characterization of the unknown phase, Li_3AlO_3

The characterization of Li_3AlO_3 was first performed by X-ray methods. Since the X-ray pattern of this compound is not available in the literature, we applied a wide range of temperature (300-600°) for the reactions of several mixtures of lithium and aluminum oxides in a Li/Al molar ratio of 3 : 1, and the X-ray spectrum of each final product was very carefully studied.

Only in the X-ray spectrum of the $Li_2O_2 + Al_2O_3$ (3 : 1) mixture, in a firing temperature range of 400-450°, were there many X-ray lines which could not be attributed to any known phase such as Li_2O , Li_2O_2 , LiOH, γ -Al₂O₃, l- and h-LiAlO₂, Li_5AlO_4 and $LiAl_5O_8$; the intensities of these lines decreased with increasing temperature, and their disappearance (for products heated above 450°) coincided with the appearance of reflections due to l-LiAlO₂ and Li_5AlO_4 . It was therefore inferred that the above unclassified X-ray lines belong to a new unknown phase, which is very unstable and only obtainable at temperatures around 400 \pm 20°.

Table 2

Product			LiOH			γ-Al ₂ O ₃			
No.	J ^a	20	d	1	20	d	I	20	d
1 2 ^b 3	vw s m	18.95 20.35 21.60	4.683 4.353 4.114	90	20.32	4.37	vw	19.5	4.55
4 5 6 7	m m vw w	22.20 23.25 28.50 29.55	4.004 3.826 3.132 3.023						
8 9 10 ^b 11	m s vs m	30.60 31.85 32.80 33.45 24.20	2.921 2.810 2.732 2.679	100	32.80	2.73	vw	32.4	2.76
12 13 ^b 14 15	m s s vw	34.20 35.65 37.00 38.15 20.00	2.622 2.515 2.430 2.359 2.250	70	35.48	2.53	m	37.2	2.42
10 17 ^b 18 19 ^c	vw vvw vvw	41.70 43.55 45.60	2.239 2.166 2.078 1.998	40	41.62	2.17	vs	45.8	1.98
20 ^ь 21 22 ^ь	vw vw m	49.15 50.35 51.40	1.857 1.812 1.774	50 75	49.25 51.33	1.85 1.78	vvw	51.1	1.79
23 ^b 24	s vvw	55.95 56.90	1.643 1.618	{70 {70	55.70 56.08	1.65 1.64			
25 26 27	vvw vw vvw	57.90 58.80 60.00	1.593 1.570 1.542						
28 29 30 ^b	w vvw m	60.45 61.40 62.40	1.531 1.510 1.488	70	62.31	1.49	vw	60.7	1.53
31 32 33° 34 ^b	VVW VVW W VW	63.60 65.60 67.00 70.00	1.463 1.423 1.397 1.343	30 30	67.92 70.24	1.38 1.34	s	67.0	1.397
35 36 37 ^b	vvw vw vvw	71.40 72.83 74.00	1.321 1.298 1.281	50	74.06	1.28			
38 ^b 39 40 41	W VVW VW VW	75.95 79.60 85.10 86.45	1.253 1.204 1.140 1.126	60 40 50 40	75.45 79.15 85.10 87.00	1.26 1.21 1.14 1.12	vvw w	78.0 87.4	1.225 1.144
42 43	VW VW	92.40 95.80	1.068 1.039	20 40	92.20 95.65	1.07			

X-ray powder spectrum of the product obtained from the reaction $Li_2O_2 + \gamma - Al_2O_3 = 1$, at 400° for 500 hrs. For comparison a list of the reflections (as taken from ref. [12] and from our reference spectra) is given for LiOH and $\gamma - Al_2O_3$

Product			LiOH			γ-Al ₂ O ₃			
No.	Ia	20	d	I	20	đ	r	20	d
44 45	vvw	99.00 100.10	1.014 1.006				vvw	97.5	1.025
46 47°	w vw	101.30 103.40	0.997 0.982	50	101.15	0.998	vvw	102.5	0.988
48 49	w vw	106,80 109.20	0.960 0.946	45	106.50	0.962			
50° 51 ^d	vw vvw	113.60 118.70	0.922 0.895	35	118.50	0.897	vvw	112.5	0.930
52 53	w vvw	120.60 124.80	0.887 0.869	50	120.50	0.888	vvw	122.0	0.881
54 55	w vw	131.60 148.60	0.844				w	145.0	0.808
56 57	w m	152.10 161.40	0.794 0.781						

Table 2 (cont.)

^a Visually estimated intensities: vs = very strong; s = strong; m = medium; w = weak; vw = very weak; vvw = very very weak.

^b Dotted lines.

^c Broad, diffuse bands.

^d Well-resolved $\alpha_1 - \alpha_2$ doublets on the film; the positions of α_1 lines were measured from line No. 51 on,

The X-ray spectrum of the product obtained by reaction of $\text{Li}_2\text{O}_2 + \text{Al}_2\text{O}_3$ at 400° for 500 hours was therefore carefully investigated. Table 2 shows that the main phases after heating consisted exclusively of the new compound and a certain amount of LiOH and Al_2O_3 . These unreacted phases were easily recognizable since those corresponding to LiOH are shown in the X-ray powder film as characteristic dotted lines whilst those belonging to Al_2O_3 appear as broad, diffuse bands.

The presence of phases such as Li_2O_2 , Li_2O_3 , Li_2O_4 , Li_5AlO_4 and $LiAl_5O_8$ in the above product was definitely excluded since their corresponding strongest X-ray lines were not revealed on the X-ray pattern, even at longer film exposures.

Table 3 reports the lines taken from Table 2, excluding those corresponding to LiOH and Al_2O_3 , and considering only those belonging to the new phase.

Moreover, the X-ray powder pattern of Li_3BO_3 , as taken from ASTM cards [12], is reported in the same Table in order to show the similarity of its spectrum with that of Li_3AIO_3 .

Analysis

The analyses for lithium and aluminum were performed by means of the atomic absorption technique, using an "Optica" instrument, and also by flame photometry (Li), using an "EEL" apparatus.

Table 3

	L	i ₃ AlO ₃		Li ₃ BO ₃ °				
No. a	1p	20	d	d	I	d	1	
1		19.06	1 692	4.50	6	1 206	3	
1	VW	18.95	4.085	4.59	15	1.360	2	
3	m	21.60	4.114	4.09	10	1.301		
4	m	22.20	4.004	3.99	10	1.330	4	
S	m	23.25	3.820	3.74	2	1.299	2	
6	VW	28.50	3.132	3.20	100	1.202	2	
7	W	29.55	3.023	3.03	100	1.189	5	
8	m	30.60	2.921	2,807	20		(
9	s	31.85	2.810	2.675	50			
11	m	33.45	2.679	2.625	22			
12	m	34.20	2.622	2.450	6		t	
14	s	37.00	2,430	2.395	4			
15	vw	38.15	2.359	2.344	40]	
16	vw v	39.90	2.259	2.315	10		ĺ	
18	vvw	43.55	2.078	2.295	10			
21	vw	50.35	1.812	2.265	2		ļ	
24	vvw	56.90	1.618	2.238	3			
25	vvw	57.90	1.593	2.214	8		l İ	
26	vw	58.80	1.570	2.069	3			
27	vvw	60.00	1.542	2.040	5	1		
28	w	60.45	1.531	1.996	10			
29	vvw	61.40	1.510	1.895	6		1	
31	vvw	63.60	1.463	1.866	25			
32	vvw	65.60	1.423	1.845	4			
35	vvw	71.40	1.321	1.777	8			
36	vw	72.85	1.298	1.758	3			
44	vvw	99.00	1.014	1.630	5	u		
45	vvw	100.10	1.006	1.608	20	[
54	w	131.60	0.844	1.591	10			
56	w	152.10	0.794	1.561	5			
57	m	161.40	0.781	1.535	6			
				1.463	4			
		i		1.440	5			
				1.408	3			
]]	

X-ray patterns of Li₃AlO₃ and Li₃BO₃

^a Numbers correspond to those in Table 2.

^b Visually estimated intensities: s = strong; m = medium; w = weak; vw = very weak; vvw = very very weak.

^c From refs. [12] and [15].

 Li_5AlO_4 (in the pure state as tested by X-ray diffraction) was completely dissolved in acidified (HCl) water and found to contain 27.9% of lithium and 21.3% of aluminum (the calculated values for the Li_5AlO_4 formula, M.W. = 125.67, are 27.61 and 21.46%, respectively). The above experimental percentages thus correspond to a Li/Al molar ratio of 5.08, in good agreement with the expected value of 5.0.

It should be remembered that the product obtained by the reaction of $\text{Li}_2\text{O}_2 + \text{Al}_2\text{O}_3$ (3 : 1) at 400° for 500 hrs contains the unknown lithium aluminum double oxide as well as some unreacted LiOH and Al_2O_3 . To find the effective Li/Al molar ratio of the new compound contained in this mixture the following procedure was followed:

a) since only the Al_2O_3 is insoluble in acidified (HCl) water, its percentage was found by weighing the residue remaining after washing repeatedly and centrifuging the powder;

b) the solution (containing the lithium aluminum double oxide and the unreacted LiOH) was analyzed and found to contain 22% and 15% by weight of Li and Al, respectively;

c) since only the LiOH was found to be soluble in ethyl alcohol, the alcoholic solution (containing the LiOH) was analyzed and found to contain 10.3% of Li.

The effective percentages contained in the lithium aluminum compound were therefore 22-10.3 = 11.7% of Li, and 15% of Al, thus giving a Li/Al molar ratio of 3.03, in good agreement with the value of 3.0 expected for the formula Li_3AlO_3 (M.W. = 95.79, calculated $\text{Li}_\% = 21.74$ and $\text{Al}_\% = 28.16$).

Thermal behaviour

For the differential thermal analysis (DTA) a Netzsch apparatus was used (DDK measuring head platinum containers, Pt-PtRh 10% thermocouples, and a heating rate of 5° /min). Heated kaolin was used as the reference material. All DTA curves were obtained using about 100 mg of undiluted samples, in an air atmosphere.



Fig. 1. a: DTA curve of the mixture $\text{Li}_2O + \gamma - \text{Al}_2O_3$ 3:1; b: DTA curve of the mixture $\text{Li}_2O + \gamma - \text{Al}_2O_3$ 5:1; c: DTA curve of Li_2O alone

It should be noted that the determinations of the heats of formation of Li_5AlO_4 and Li_3AlO_3 were performed by the DTA method since: a) the compounds are very hygroscopic and this could lead to large errors in the measurements of ΔH_0 by the usual solution calorimeters; and b) after detection of the peaks by the DTA method, an immediate X-ray test on the powder can check that the above compounds were really formed.



Fig. 2. a: DTA curve of the mixture $Li_2O_2 + \gamma - Al_2O_3 \ 3:1$; b: DTA curve of the mixture $Li_2O_2 + \gamma - Al_2O_3 \ 5:1$; c: DTA curve of Li_2O_2 alone

The DTA curves corresponding to the $Li_2O + Al_2O_3$ mixtures in the ratios 3 : 1 and 5 : 1 show (Fig. 1a and 1b) only one sharp exothermic peak in the range $450-530^\circ$, due to the formation of the compound Li_5AlO_4 , as confirmed by the X-ray pattern.

The DTA curves corresponding to the $\text{Li}_2\text{O}_2 + \text{Al}_2\text{O}_3$ mixtures in the ratios 3 : 1 and 5 : 1 (Figs 2a and 2b) show an endothermic peak at $320-380^\circ$ due to the decomposition of Li_2O_2 into Li_2O and $1/2 \text{O}_2$ (confirmed by the DTA curve of pure Li_2O_2 , Fig. 2c), and two exothermic peaks in the ranges $400-440^\circ$ and $450-510^\circ$, respectively. The first one is due to the formation of Li_3AlO_3 , the second to that of Li_5AlO_4 .

The appearance of only one exothermic peak in the range $450-530^{\circ}$ (Fig. 1a and 1b) by mixing Li₂O and Al₂O₃ in Li/Al molar ratios 3 : 1 or 5 : 1 shows that under these conditions only Li₅AlO₄ is formed. The presence of a weak endothermic peak at $400-425^{\circ}$ in these curves led us to suppose that, under the described operational conditions of sampling, Li₂O did not remain uncontaminated, and this was not surprising considering the well-known ready tendency of Li₂O

to react with water and carbon dioxide in air. This supposition was further confirmed by the DTA curve of Li₂O alone (Fig. 1c), and from its X-ray spectrum; since the latter showed the presence of LiOH and Li₂CO₃ (in small quantities) together with the main Li₂O phase, the two endothermic peaks in the range $400-500^{\circ}$ of the DTA curve in Fig. 1c are therefore due to the melting of LiOH (the DTA curve of LiOH alone shows only one endothermic reaction at $400-430^{\circ}$, exothermic in the cooling curve in the range $400-380^{\circ}$) and to the initial decomposition of Li₂CO₃ (which in fact starts to decompose over 440° , respectively [11]).

It is therefore reasonable to infer that the presence of surface LiOH and Li_2CO_3 hinders the reactivity of Li_2O with Al_2O_3 up to 450° and thus prevents the formation of Li_3AlO_3 . Only at 450° can uncontaminated Li_2O react with Al_2O_3 to give Li_5AlO_4 and l-LiAlO₂ (at Li/Al = 3 : 1), or Li_5AlO_4 alone (at Li/Al = 5 : 1), as observed by X-ray powder spectra.

In contrast, when Li_2O_2 is used as reactant (Fig. 2) two exothermic peaks are always observed; the peak at 400-440° is more consistent if the Li/AI ratio is 3 : 1. It is evident that in this case the conditions are more favourable for the formation of Li₃AlO₃, and that this compound is an intermediate in the Li₅AlO₄ formation from 5 : 1 mixtures. It may be assumed that Li₃AlO₃ is formed only by Li₂O₂ because the temperature of complete decomposition of Li₂O₂ to give $\text{Li}_2\text{O} + \frac{1}{2}$ O₂ almost coincides with the temperature of formation of Li₃AlO₃, and Li₂O₂ is thus a sure source of uncontaminated Li₂O. It should be noted that the smooth and large endothermic peaks observed in all the DTA curves of Li₂O or Li₂O₂ + Al₂O₃ in the temperature range 120-250° are probably due to the dehydration of Al₂O₃, which was believed to contain 10% water.

The experimental determination of the heats of formation of the compounds synthetized here was based on the direct calorimetric measurement of the heats of reactions in the solid state from the constituents, the heats of formation of the starting oxides being known from the literature in the temperature range where the reactions

 $3 \text{ Li}_{2}\text{O}(S) + \text{Al}_{2}\text{O}_{3}(S) = 2 \text{ Li}_{3}\text{AlO}_{3}(S) \qquad (1) \text{ at } 400^{\circ}$ $5 \text{ Li}_{2}\text{O}(S) + \text{Al}_{2}\text{O}_{3}(S) = 2 \text{ Li}_{5}\text{AlO}_{4}(S) \qquad (2) \text{ at } 530^{\circ}$

 $5 \operatorname{Li}_{2}O(5) + \operatorname{Ai}_{2}O_{3}(5) - 2 \operatorname{Li}_{5}\operatorname{Ai}O_{4}(5)$ (2) at 550

occur. The values of the heats of reaction $\Delta H(1)$ and $\Delta H(2)$ were obtained from the areas of the well-defined exothermic peaks observed in the DTA curves.

The apparatus was previously calibrated in the same temperature range with compounds of known heats of fusion, such as $CsNO_3$ and $K_2Cr_2O_7$.

For Li₃AlO₃ the heating of Li₂O₂ + Al₂O₃ mixtures in a molar ratio of 3 : 1 was halted at 440°, after the formation of the product and before the start of its decomposition into Li₅AlO₄ and LiAlO₂. The amount of Li₃AlO₃ formed was derived from the weight of unreacted Al₂O₃; in fact the amount of alumina not dissolved in HCl (1 : 8) with respect to the alumina initially weighed gives the

amount of product formed. In the calculation it was also considered that the Al_2O_3 used contained (as observed experimentally) about 10% water.

The heat of reaction (1):

$$3 \operatorname{Li}_2 O + \operatorname{Al}_2 O_3 = 2 \operatorname{Li}_3 \operatorname{Al} O_3$$

averaged from four different runs, was found to be -16.5 ± 2 kcal/mole. By using the heats of formation at 400° of Li₂O (-138.55 kcal/mole) [13] and of γ -Al₂O₃ (-384.84 kcal/mole) [14], the measured heat of reaction (1), -16.5 ± 2 kcal/mole, corresponds to a calculated heat of formation of Li₃AlO₃ of -416.75 ± 2 kcal/mole at 400°.

The determination of the heat of formation of Li_5AlO_4 was performed in the same way by weighing the area of the exothermic peak obtained from the reaction of $\text{Li}_2\text{O} + \text{Al}_2\text{O}_3$ (5 : 1) at 470°.

The heat of reaction (2):

$$5 \operatorname{Li}_2 O + \operatorname{Al}_2 O_3 = 2 \operatorname{Li}_5 \operatorname{Al} O_4$$

was in quite good agreement irrespective of the lithium oxide used and was found to be -13.5 ± 0.8 kcal/mole (average of five runs with both types of mixture). By using the same values of the heats of formation of Li₂O and γ -Al₂O₃ as for reaction (1), the measured heat of reaction (2), -13.5 ± 0.8 kcal/mole, leads to a calculated heat of formation of Li₅AlO₄ of -552.3 ± 0.8 kcal/mole at 500°.

Discussion

The results of our investigation clearly indicate that the formation of some lithium aluminum oxides, such as Li_3AIO_3 and Li_5AIO_4 , depends on a very limited range of physical variables. As usual when reactivity conditions are investigated, we have varied the type of starting materials, the composition range of the reagents, as well as the temperature of reaction.

It has been found that Li_5AIO_4 may be prepared in a pure form by calcining a mixture of Li_2O_2 and γ -Al_2O_3 in a molar ratio of 5 : 1 at 500° in air for several hours (140), as well as by reaction between $Li_2O + Al_2O_3$. In contrast, if Li_2CO_3 or LiNO₃ are used as starting lithium reagents, phases other than Li_5AIO_4 are formed. The Li_5AIO_4 phase is stable in the temperature range 450-650°; above 650° small amounts of other phases, such as 1-LiAIO₂, are revealed by X-ray spectra. This range of stability was observed when lithium oxides were used as reagents; when Li_2CO_3 or LiNO₃ are used the upper limit of stability is lowered.

However, it is much more difficult to prepare Li_3AlO_3 ; until now, in fact, this compound was unknown, its existence only having been postulated by a few authors [9, 10]. We have found that it can be formed during the reaction in the solid state between Li_2O (obtained during the decomposition of the initial

 Li_2O_2) and γ -Al₂O₃ (a very reactive form of alumina due to its high surface area), with a Li/Al molar ratio of 3 : 1, and in a very narrow range of temperature, namely at 400 ± 20° in air; above this temperature it decomposes giving the meta-aluminate, l-LiAlO₂ and Li₅AlO₄. Moreover, it is formed only by reaction of lithium peroxide with alumina, since attempts to prepare it from lithium oxide, lithium carbonate or lithium nitrate were unsuccessful.

The formation of Li_3AlO_3 by the reaction in the solid state between Li_2O_2 and Al_2O_3 (3 : 1) at 400° for 500 hrs has been confirmed by chemical analysis and X-ray spectra. However, both chemical analyses and X-ray spectra have shown that the reaction is not complete, since the powder, even after prolonged heating (500 hrs), still contains a certain amount (about 40%) of unreacted materials such as LiOH and Al_2O_3 .

It is very interesting to note that the spectrum of this compound is very similar to that of Li_3BO_3 [15] (one of the two types reported in the literature for this compound [12]). Table 3 shows a clear correlation between the first lines of Li_3AIO_3 characterized here and of Li_3BO_3 studied by Sastry and Hummel [15]; the *d* values show a special agreement with one another and it is therefore suggested that the two compounds are almost similar with respect to their crystal symmetries, unit cells and structures. The two compounds have very similar formulas, and should give, if they are isomorphous (as the similarity of their X-ray spectra suggests), the formation of solid solutions.

It should finally be noted that the temperature of formation found here for Li_3AlO_3 is rather different from that observed (500°) by Kröger and Fingas [9].

The heats of formation of Li_3AlO_3 and Li_5AlO_4 have been calculated from DTA curves, as stated above, to be -416.75 ± 2 and -552 ± 0.8 kcal/mole, respectively.

The DTA curves have also confirmed the observations from the numerous attempts at calcination, and have made it possible to define the stabilities of the various phases formed during the reaction, and to understand the different reactivities with respect to Al_2O_3 of the several lithium compounds used in the preparation.

References

- 1. L. DEBRAY and A. HARDY, Compt. Rend. Acad. Sci. Paris, 251 (1960) 725.
- 2. P. HAGENMULLER and L. DEBRAY, ibid., 250 (1960) 134.
- 3. E. KORDES, Z. Krist., 91 (1935) 193.
- 4. H. A. LEHMANN and H. HESSELBARTH, Z. anorg. allg. Chem., 313 (1961) 117.
- 5. E. ZINTL and W. MORAWIETZ, ibid., 236 (1938) 372.
- 6. F. A. HUMMEL, B. S. R. SASTRY and D. WOTRING, J. Am. Ceram. Soc., 41 (1958) 88.
- 7. H. A. LEHMANN and H. HESSELBARTH, Z. anorg. allg. Chem., 315 (1962) 14.
- 8. K. TORKAR, H. P. FRITZER and H. KRISCHNER, Science of Ceramics, 2 (1962) 19.
- 9. C. KRÖGER and E. FINGAS, Z. anorg. allg. Chem., 224 (1935) 289.
- 10. T. F. FEDOROV and F. I. SHAMRAI, Prim. Vak. v. Met., Akad. Nauk SSSR, Inst. Met. im A. A. Baikova, 137 (1960).

- 11. C. DUVAL, "Inorganic Thermogravimetric Analysis". Elsevier Publ., 1963, pp. 166-167. 12, ASTM X-ray Powder Data File, Baltimore, Md., 1957.
- 13. C. E. WICKS and F. E. BLOCK, Thermodynamic Properties of 65 Elements. Bureau of Mines, Bull. no. 605.
- 14. Handbook of Chemistry and Physics, 49th Edition, 1968. The Chem. Rubber Co., Cleveland, Ohio, p. D 38 and following.
- 15. B. S. R. SASTRY and F. A. HUMMEL, J. Am. Ceram. Soc., 42 (1959) 216.

RÉSUMÉ – On a recherché les meilleures conditions pour réaliser la synthèse à l'état solide des oxydes doubles Li_5AlO_3 et Li_3AlO_3 en partant des oxydes individuels et pour déterminer leur chaleur de formation. Li_5AlO_4 a été obtenu en partant de Li_2O_2 (ou Li_2O) et de γ -Al₂O₃ dans le rapport molaire Li/Al = 5/1; il a été caractérisé par étude aux rayons X. L'orthoaluminate de lithium, Li_3AlO_3 , a été obtenu en partant de Li_2O_2 et de γ -Al₂O₃ mélangés dans le rapport 3/1. L'analyse chimique a confirmé la formule présumée Li_3AlO_3 . L'ATD a permis de déterminer le domaine de stabilité thermique de ces composés: très restreint pour Li_3AlO_3 (400–430°), plus grand pour Li_5AlO_4 (440–plus de 600°), ainsi que leurs chaleurs de formation: -416.8 ± 2 kcal/mole et -552.3 ± 0.8 kcal/mole, respectivement.

ZUSAMMENFASSUNG – Es wurden die Bildungsverhältnisse der doppelten Oxide Li_5AlO_4 und Li_3AlO_3 in fester Phase ausgehend von den einfachen Oxyden geprüft und ihre Bildungswärmen bestimmt. Li_5AlO_4 wurde aus Li_2O_2 oder Li_2O und γ -Al₂O₃ beim Molverhältnis von Li/Al 5 : 1 erhalten und röntgenographisch identifiziert. Das Orthoaluminat Li_3AlO_3 erhielt man beim Molverhältnis 3 : 1 von Li_2O_2 und γ -Al₂O₃. Die Zusammensetzung von Li_3AlO_3 wurde durch chemische Analyse nachgewiesen. Die DTA-Prüfung zeigte, daß Li_3AlO_3 nur im sehr kleinen Temperaturgebiet (400–430°), Li_5AlO_4 hingegen im weiteren Bereich zwischen 440–600° stabil ist. Die Bildungswärmen betrugen für $Li_5AlO_4 - 552.3 \pm 0.8$, für $Li_3AlO_3 - 416.8 \pm 2$ kcal/Mol.

Резюме — Цель исследования — поиск наилучших условий синтеза двойных окисей Li₅AlO₄ и Li₃AlO₃ в твердой фазе на основе простых окисей и определение теплот их образования. Li₅AlO₄ получен из Li₂O₂ или Li₂O и γ -Al₂O₃ при молярном соотношении Li/Al 5 : 1, и его структура подтверждена рентгендифракционным методом. Ортоалюминат лития, Li₃AlO₃, получен из Li₂O₂ и γ -Al₂O₃ при молярном соотношении 3 : 1. Предполагаемая формула подтверждена химическим анализом. Методом ДТА установлено, что область температур, в которой эти соединения стабильны, довольно ограничена для Li₃AlO₃ (400—430 °C) и более широкая для Li₅AlO₄ (440— \leq 600 °C). Теплоты образования Li₅AlO₄ и Li₃AlO₃ также определены с помощью метода ДТА и составляют 552.3 \pm 0.8 и 416.8 \pm 2 Ккал/мол, соответственно.